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Effective Nano-manufacturing of T-Nb2O⁵ for supercapacitor applications

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Abstract: Characterized by unique physical and chemical properties, metal oxide materials have garnered significant attention for research and development in energy storage device applications. In the current work, we present a simple and low-cost synthesis protocol for orthorhombic-phase niobium oxide $(T-Nb₂O₅)$ electrodes, aimed at supercapacitor applications. The as-prepared T-Nb₂O₅ was characterized utilizing field emission scanning electron microscopy, X-ray diffraction, and energy-dispersive X-ray spectroscopy, confirming the formation of orthorhombic-phase T-Nb₂O₅ nanoparticles. Detailed electrochemical analyses were conducted on $T-Nb₂O₅$, utilizing 1 M LiOH as the electrolyte. The unique nanoparticle architecture of $T-Nb_2O_5$ offers abundant electro-active sites and enhances reaction kinetics, leading to high specific capacitance. Notably, the T-Nb₂O₅ electrode achieved a gravimetric capacitance of approximately 23 F g^{-1} at the lowest sweep rate (5 mV s⁻¹). These findings highlight the potential of $T-Nb_2O_5$ as an effective electroactive material for supercapacitors.

Keywords: hydrothermal; T-Nb₂O₅; nanoparticles; specific capacitance; supercapacitors

1. Introduction

In the present age, electrochemical energy storage devices are key for tackling non-renewable energy source depletion and reducing the impact of global warming. Amid these devices, lithium-ion batteries (LiBs) and supercapacitors have attracted significant attention due to their widespread industrial and daily applications, owing to their high-specific power, rapid charge-discharge rates, and prolonged cyclic stability [1–3]. Supercapacitors, comprising electrochemical double-layer capacitors (EDLCs) and pseudocapacitors, provide the benefit of rapid energy release via fast surface or near-surface electrochemical reactions, including physical adsorption/desorption or Faradaic processes [4]. In both EDLCs and pseudo capacitors, carbon-based materials, transition metal oxides, and transition metal hydroxides are widely used as electroactive materials in academic research and industrial applications. Transition metal oxides are considered an aspiring candidate for use as electrodes in energy storage devices due to their abundant availability, ecofriendliness, and ease of accessibility [5]. These materials also possess a variety of attractive features, including diverse compositions and morphologies, large surface areas, and high theoretical gravimetric specific capacitance. Moreover, transition metal oxides are crucial in the electrodes of electrochemical supercapacitors, as they significantly enhance capacitance by enabling precise adjustment and control of defects and surface/interfaces at the nanoscale [6].

Metal oxides, including cobalt oxide, iron oxide, nickel oxide, and manganese oxide, serve as advantageous electrodes for supercapacitors due to their unique chemical and physical properties, as well as their high specific capacity and

capacitance, which substantially surpass those of carbon-based electrodes. Among various metal oxides, $Nb₂O₅$ exhibits various stoichiometries and crystal structures that depend on the synthesis method and has been extensively studied as a highcapacity material for energy storage applications [7–11]. In the last few years, T- $Nb₂O₅$ has emerged as a promising anode material for $Na⁺$ and $Li⁺$ storage due to its large interplanar lattice spacing along the (001) plane and its characteristic pseudocapacitive behavior [12–19]. However, there are limited reports on the use of $T-Nb₂O₅$ as an electrode material for supercapacitors. At first, Kong et al. [20] reported free- standing $T-Nb_2O_5/graphene$ composite papers for Li-intercalating pseudocapacitive electrodes. The T-Nb₂O₅/graphene composite paper obtained a gravimetric and volumetric capacitance of 620.5 F g^{-1} and 961.8 F cm⁻³ at 1 mV s⁻¹.

Later on, Jiang et al. $[21]$ synthesized T-Nb₂O₅/N-doped carbon nanosheets for use in lithium-ion capacitors. These lithium-ion capacitors, based on $T-Nb₂O₅/N$ doped carbon nanosheets, achieved an energy density of approximately 70.3 Wh kg−1 and a power density of 16,014 W kg⁻¹. Furthermore, Zhang and their colleagues [22] prepared $T-Nb₂O₅$ nanoparticles confined within a porous carbon shell for use in a Hybrid supercapacitor, achieving a maximum specific capacity of 410 Fg^{-1} at a Current density of 1Ag−1 . These electrochemical studies suggest that additional research is needed to further study the electrochemical behavior of bare $T-Nb₂O₅$ as an electroactive material for energy storage systems.

Therefore, in the current work, we synthesized bare $T-Nb₂O₅$ utilizing the hydrothermal method and conducted electrochemical evaluations as supercapacitor electrodes, using both half-cell and symmetric supercapacitor devices.

2. Experimental methods

2.1. Materials

Niobium(V) chloride [NbCl₅], Ethanol [C₂H₅OH], and Lithium hydroxide [LiOH] were purchased from Sigma Aldrich, India. Polyvinylidene fluoride (PVDF) and Carbon black were obtained from Sigma Aldrich, India.

2.2. Growth of T-Nb2O⁵ nanostructure

All chemical reagents were directly used as purchased without further purification. A highly reproducible hydrothermal method was used to prepare a uniform amorphous precursor. In a typical method, 2 mmol of NbCl₅ was dissolved in 50 mL of ethanol, resulting in a yellow solution, which was stirred for 30 min until it became a colorless solution. After adding 20 mL of deionized water and stirring for 2 h, the opaque sol was transferred to a Teflon autoclave and sealed within a steel container. The autoclave was then heated to $200\degree C$ for 12 h and subsequently cooled naturally. Once the reaction was complete, the precipitates were extracted from the autoclave and washed several times with water and ethanol to remove any residues. The collected powder was then dried overnight at 60 °C. Finally, the dried sample was annealed at 600 °C for 2 h.

2.3. Physical characterization

X-ray diffraction (XRD) analysis of the as-prepared $T-Nb₂O₅$ was conducted utilizing an XRD instrument bought from Malvern Panalytical equipped with Cu-Kα radiation ($\lambda = 1.54184$ Å), operating at a high voltage of 30 kV and a current of 30 mA. The surface morphology of the prepared $T-Nb₂O₅$ nanostructure was examined utilizing FESEM (JEOL JSM 7900F) at various magnifications.

2.4. Electrochemical characterization

The electrode material was prepared utilizing the slurry coating technique. In brief, the electroactive material $(T-Nb₂O₅)$, PVDF and carbon black were mixed in a weight ratio of 80:5:15 with N-methyl pyrrolidone (NMP) as the dispersant. The mixture was ground for a few hours in an agate mortar to make a uniform slurry. The slurry was applied onto a stainless-steel substrate and dried at 55 °C for 8 h.

The electrochemical characterization of $T-Nb₂O₅$ was carried out utilizing a three-electrode system in a 1.0 M LiOH electrolyte, where Ag/AgCl was used as the reference electrode, platinum foil as the counter electrode, and $T-Nb₂O₅$ -coated stainless steel as the working electrode.

A T-Nb₂O₅//T-Nb₂O₅ symmetric supercapacitor device (SSD) was constructed by inserting a polypropylene separator between two stainless steel electrodes coated with T-Nb₂O₅. The electrochemical performance of both the T-Nb₂O₅ electrode and the T-Nb₂O₅//T-Nb₂O₅ SSD was evaluated through charge-discharge (CD), cyclic voltammetry (CV), and electrochemical impedance spectroscopy (EIS) utilizing an electrochemical workstation. The specific capacitance of the $T-Nb₂O₅$ electrodes was calculated using the following Equations (1) and (2):

$$
Specific\ capacitance = (IdV)/(s \times AV \times m)
$$
 (1)

$$
Specific \; capacitance = (I \times \Delta t)/(\Delta V \times m) \tag{2}
$$

here, " I " is the current (A), " ΔV " is the potential window, specific capacitance in $F g^{-1}$, "*s*" is the scan rate (mV s⁻¹), " Δt " is the discharge time (s), and "*m*" is the mass ofthe electroactive material coated on the substrate.

3. Results and discussion

The $T-Nb₂O₅$ was synthesized using a hydrothermal method, followed by postannealing. The XRD pattern of $T-Nb₂O₅$ is illustrated in **Figure 1a**, which indicates that the crystal structure of pure $Nb₂O₅$ aligns closely with the standard peaks for T-Nb₂O₅ (JCPDS No. 30–0873) [23]. The peaks at 22.79°, 28.50°, 36.83°, 42.88°, 45.31°, 46.47°, 50.06°, 51.04°, 55.43°, 56.57°, 59.02°, 63.93°, 71.26°, and 77.95° correspond the (001), (100), (181), (130), (110), (002), (301), (331), (182), (381), (160), (161), (382) and (122). To examine the microstructure and morphology of the assynthesized sample, scanning electron microscope (SEM) analysis was conducted, as presented in **Figure 1b** and **Figure 1c**. The SEM micrograph discloses the growth of $Nb₂O₅$ nanoparticles with an average size of around 30–40 nm. Additionally, EDS analysis (as presented in **Figure 1d**) was performed to confirm the presence of niobium and oxygen elements in the sample. The overlay elemental mapping of

Nb2O⁵ nanoparticles, shown in **Figure 1e** and **Figure 1f**, demonstrates the uniform distribution of niobium and oxygen elements throughout the sample [24].

Figure 1. (a) XRD pattern of as-prepared T-Nb₂O₅. FE-SEM micrographs of T-Nb₂O₅; (b–c) at various magnifications (200.0 kx, and 150.0 kx); **(d)** the EDS spectrum of as-prepared T-Nb₂O₅. The elemental mapping of T-Nb2O5; **(e)** niobium; **(f)** oxygen element.

To evaluate the supercapacitive performance of the $T-Nb₂O₅$ nanoparticles, cyclic voltammetry (CV) tests and galvanostatic charge-discharge (CD) cycling were conducted in a three-electrode system. The CV profile of $T-Nb₂O₅$ was conducted (as presented in **Figure 2a**) in the potential range of −1.0 to 0.2 V at various scan rates of 5 to 100 mV s⁻¹. The CV profile illustrates that it is different from EDLC-based material. It can be perceived that strong redox peaks are noticeable in each CV profile, signifying that the measured electrochemical performance is mostly founded on the redox mechanism or pseudocapacitive behavior of $Nb₂O₅$ [25]. The anodic and cathodic peaks in the CV profile appear at 0.35 V and 0.50 V, respectively.

Figure 2. Electrochemical characterization of T-Nb₂O₅ electrode in a three-electrode system. (a) CV curves of T-Nb₂O₅ electrode at various scan rates $(5-100 \text{ mV s}^{-1})$; (b) CD profile of T-Nb₂O₅ electrode at various applied currents $(1-5 \text{ mA})$; **(c)** effect of scan rates on specific capacitance of $T\text{-Nb}_2O_5$ electrode; and **(d)** the Nyquist plot $T\text{-}$ $Nb₂O₅$ electrode with inset shows the enlarged view.

These peaks are attributed to the intercalation and deintercalation of lithium ions at the surface of the $T-Nb₂O₅$ electrode, indicating its pseudocapacitive behavior. The Charge storage mechanism, involving lithium-ion intercalation $T-Nb₂O₅$, can be represented by the following reaction [26]:

$$
Nb_2O_5 + xLi^+ + xe^- \leftrightarrow Li_xNb_2O_5
$$

The increase in current with corresponding increases in scan rates in the CV profiles suggests the capacitive nature of the $T-Nb₂O₅$ electrode. Furthermore, the CD profile of the T-Nb₂O₅ electrode at various applied currents (1 to 5 mA) is provided in **Figure 2b**. The CD profiles indicate the distinct plateau regions in the discharge curves, which again demonstrate the pseudocapacitive nature of the $T-Nb₂O₅$ electrode [27]. In the CD profile, the intermediate resistance (IR) drop arises primarily from the Internal resistance within the active electrode material, along with contact resistance at the electrode-electrolyte interface. Notably, the observed IR drop gradually decreases as the current is reduced [28,29]. As shown in **Figure 2b**, the measured IR drop for the T-Nb₂O₅-based electrode at a constant current of 3 mA is approximately 0.08 V. The scan rate versus the specific capacitance curve of the T- $Nb₂O₅$ electrode is provided in **Figure 2c**. It showed that the specific capacitance of the T-Nb₂O₅ electrode is increased from 14 to 23 F g^{-1} with a decrease in scan rate from 100 to 5 mV s−1 . The Electrochemical impedance spectroscopy (EIS) measurements were conducted for the $T-Nb₂O₅$ electrode over a frequency range of

0.1 Hz to 100 kHz to investigate its charge transfer behavior, as illustrated in **Figure 2d**. The enlarged view of the Nyquist plot reveals an equivalent series resistance (ESR) of approximately 0.5 Ω, with no noticeable charge transfer resistance, indicating the excellent electrical conductivity of the $T-Nb₂O₅$ electrode.

To further evaluate the electrode composed of $T-Nb₂O₅$ nanoparticles, a symmetric supercapacitor device (SSD) was fabricated, employing $T-Nb_2O_5$ electrodes as both the positive and negative electrodes. **Figure 3a** presents the CV profile of the T-Nb₂O₅-based SSD, demonstrating its operation within a potential range of 0.0 to 1.0 V. Furthermore, in the CV profile, no noticeable distortion is observed as the scan rates increase, indicating the rapid intercalation reaction of the T-Nb₂O₅ electrode. The CD tests for the T-Nb₂O₅-based SSD, conducted at various applied currents, are displayed in **Figure** 3b. The CD profile of T-Nb₂O₅-based SSD exhibits a quasi-rectangular behaviour, which aligns well with the CV results.

Figure 3. Electrochemical characterization of T-Nb₂O₅ in an SSD system. **(a)** CV profile of T-Nb₂O₅-based SSD at various scan rates $(5-100 \text{ mV s}^{-1})$; **(b)** chargedischarge-profile of T-Nb₂O₅ based SSD at various applied currents $(0.1-1 \text{ mA})$; **(c)** effect of scan rates on specific Capacitance of $T-Nb₂O₅$ based SSD; and **(d)** the Nyquist plot $T-Nb₂O₅$ based SSD with inset shows the enlarged view.

Figure 3c illustrates the relationship between specific capacitance and scan rate for the T-Nb₂O₅-based SSD, which achieved a specific capacitance of approximately 0.76 F g^{-1} at a scan rate of 5 mV s⁻¹. The cyclic stability test of the fabricated T- $Nb₂O₅$ -based SSD over 3000 cycles at an applied constant current of 0.5 mA, as illustrated in **Figure** S1 (in supporting information). The $T-Nb₂O₅$ -based SSD demonstrated a capacitance retention of approximately 84.23%, with a calculated

coulombic efficiency of around 91% over continuous 3000 cycles. The Nyquist plot for T-Nb₂O₅-based SSD is represented in **Figure 3d**. The Nyquist plots for T-Nb₂O₅based SSD displayed three distinguishing regions: (i) low-frequency, (ii) intermediatefrequency, and (iii) high-frequency. These regions allowed for determining key parameters such as the knee frequency, Warburg line, an equivalent series resistance (ESR) of the devices [30]. The T-Nb₂O₅-based SSD obtained a solution resistance (Rs) of about 0.7 Ω and charge transfer resistance (Rct) of about 2.3 Ω . The fitted Randles circuit of the Nyquist plot is provided in **Figure S2** (in supporting information). Further, the EIS measurements of the $T-Nb₂O₅$ -based SSD before and after the cyclic stability test, are illustrated in **Figure S3** (in Supporting information). The Nyquist plots indicate a change in solution resistance (Rs) from 0.7 Ω to 0.9 Ω and an increase in charge transfer resistance (Rct) from 2.3 Ω to 3 Ω after the cyclic stability test over 3000 cycles.

4. Conclusions

In summary, a simple and efficient technique was employed to synthesize orthorhombic-phase niobium oxide $(T-Nb₂O₅)$.

The synthesized $T-Nb₂O₅$ was then characterized utilizing a range of physical techniques, including XRD, SEM, and EDS.

Additionally, the supercapacitive performance of the $T-Nb₂O₅$ electrode was evaluated using both a half-cell system and a symmetric supercapacitor device.

The $T-Nb₂O₅$ electrode demonstrated a gravimetric capacitance of approximately 23 F g^{-1} at a scan rate of 5 mV s⁻¹, along with excellent rate capability.

These findings highlight the potential of $T-Nb₂O₅$ nanoparticles for future generation energy storage systems.

Supplementary materials: The supplementary material includes the cyclic stability test, the fitted Randles circuit, and Nyquist plots (both before and after the cyclic stability test) for T-Nb₂O₅-based SSD.

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Conflict of interest: The authors declare no conflict of interest.

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